

Reorientacja spinowa i kontrola anizotropii magnetycznej w niskowymiarowych układach ferro- i antyferromagnetycznych

WFiIS, 8.03.2019

dr inż. Michał Ślęzak
Katedra Fizyki Ciała Stałego
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1. X-ray photoemission electron microscopy study of the in-plane spin reorientation transitions in epitaxial Fe films on W(110).
M. Ślęzak, T. Giela, D. Wilgocka-Ślęzak, A. Koziół-Rachwał, T. Ślęzak, R. Zdyb, N. Spiridis,
C. Quitmann, J. Raabe, N. Pilet, J. Korecki
Journal of Magnetism and Magnetic Materials 348 (2013) 101–106

2. Prospects of X-ray photoemission electron microscopy at the first beamline of the Polish synchrotron facility SOLARIS.
M. Ślęzak, T. Giela, D. Wilgocka-Ślęzak, N. Spiridis, T. Ślęzak, M. Zając, A. Koziół-Rachwał,
R.P. Socha, M. Stankiewicz, P. Warnicke, N. Pilet, J. Raabe, C. Quitmann, J. Korecki
X-Ray Spectrometry 44 (2015) 317–322

3. Giant in-plane magnetic anisotropy in epitaxial bcc Co/Fe(110) bilayers
M. Ślęzak, T. Ślęzak, K. Matlak, B. Matlak, P. Drózdź, T. Giela, D. Wilgocka-Ślęzak, N. Pilet, J. Raabe, A. Koziół-Rachwał, J. Korecki,
Physical Review B 94 (2016) 014402

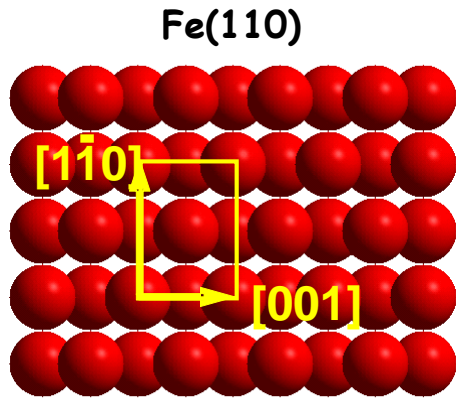
4. Adsorption induced modification of in-plane magnetic anisotropy in epitaxial Co and Fe/Co films on Fe(110)
M. Ślęzak, T. Ślęzak, K. Matlak, P. Drózdź, J. Korecki
AIP Advances 8 (2018) 056806

5. Multiple spin reorientation transitions and large in plane magnetic anisotropy in epitaxial Au/Co/Fe(110) films
M. Ślęzak, P. Drózdź, K. Matlak, A. Koziół-Rachwał, J. Korecki, T. Ślęzak
Journal of Magnetism and Magnetic Materials 475 (2019) 195–200

6. How a ferromagnet drives an antiferromagnet in exchange biased CoO/Fe(110) bilayers
M. Ślęzak, T. Ślęzak, P. Drózdź, B. Matlak, K. Matlak, A. Koziół-Rachwał, M. Zając, J. Korecki
Scientific Reports 9 (2019) 889

Introduction

Spin Reorientation Transition in Fe/W(110)



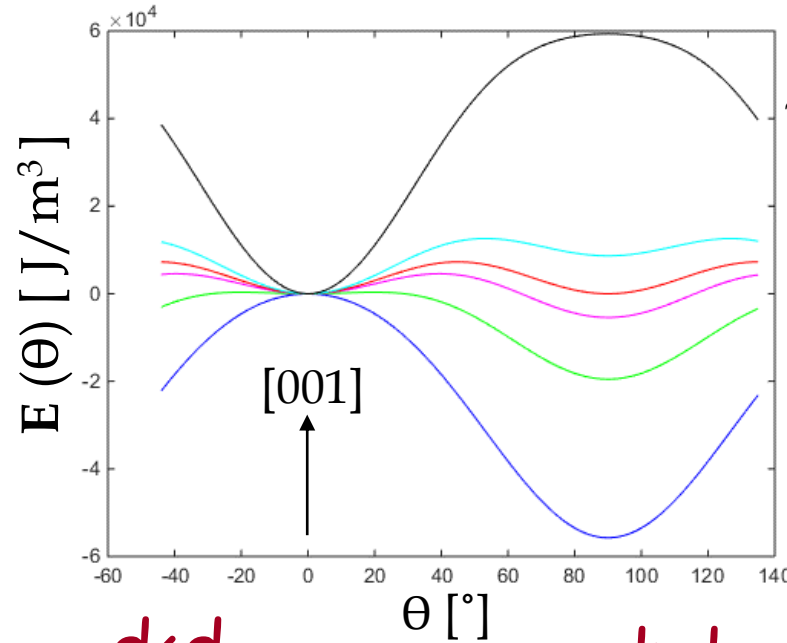
d_{crt} is a good measure of in-plane magnetic anisotropy

$$E_{\text{m in-plane}}(\theta) = A \sin^2(\theta) + B \sin^4(\theta)$$

$$A = K_{\text{vp}} - K_{\text{sp}}/d$$

$$B = K_{\text{vpp}} - K_{\text{spp}}/d$$

[1-10] anisotropy



~ bulk Fe

150 Å

130 Å - critical SRT thickness

120 Å

100 Å

70 Å

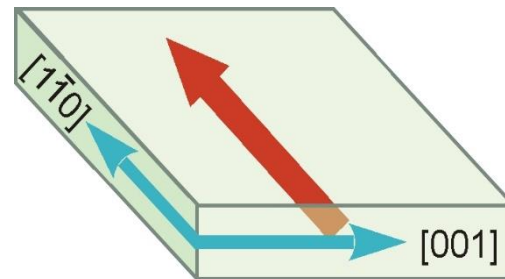
$$K_{\text{vp}} = 10.60 \cdot 10^4 \text{ J/m}^3$$

$$K_{\text{vpp}} = -0.60 \cdot 10^4 \text{ J/m}^3$$

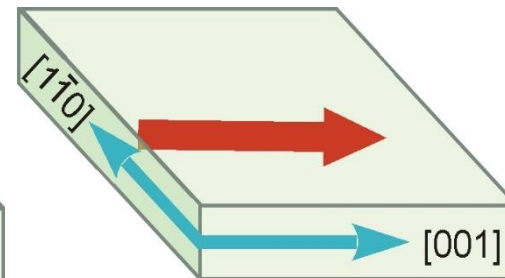
$$K_{\text{sp}} = 0.82 \cdot \text{mJ/m}^2$$

$$K_{\text{spp}} = 0.48 \cdot \text{mJ/m}^2$$

$d < d_{\text{crt}}$



$d > d_{\text{crt}}$



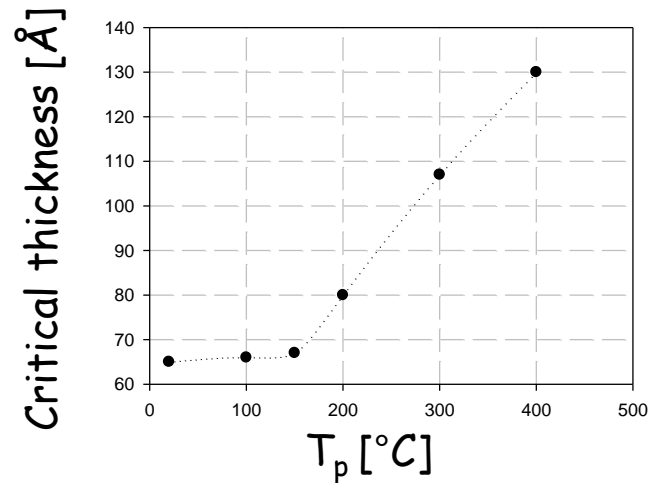


**FM/FM:
Co/Fe(110)**

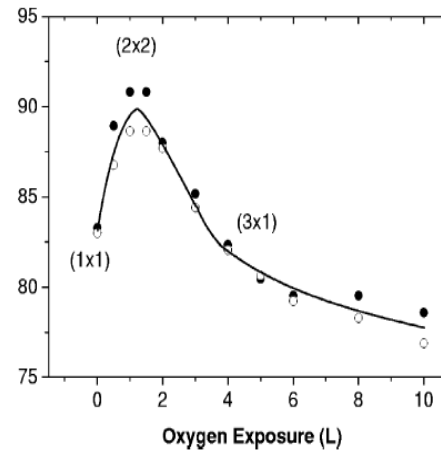
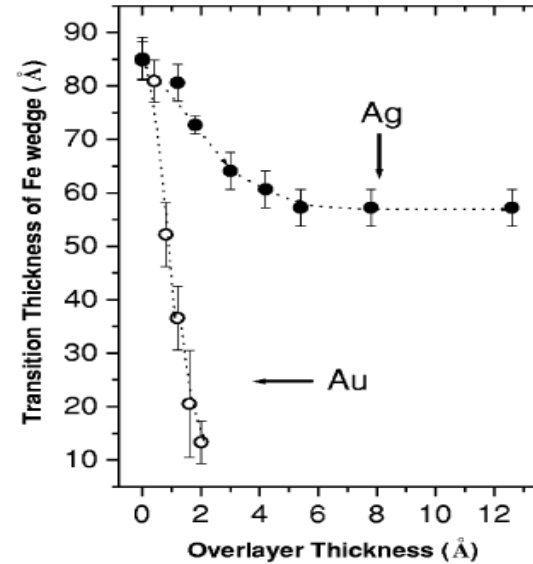


**AFM/FM:
CoO/Fe(110)**

Introduction to part I: modifications of magnetic anisotropy and SRT in Fe/W(110)

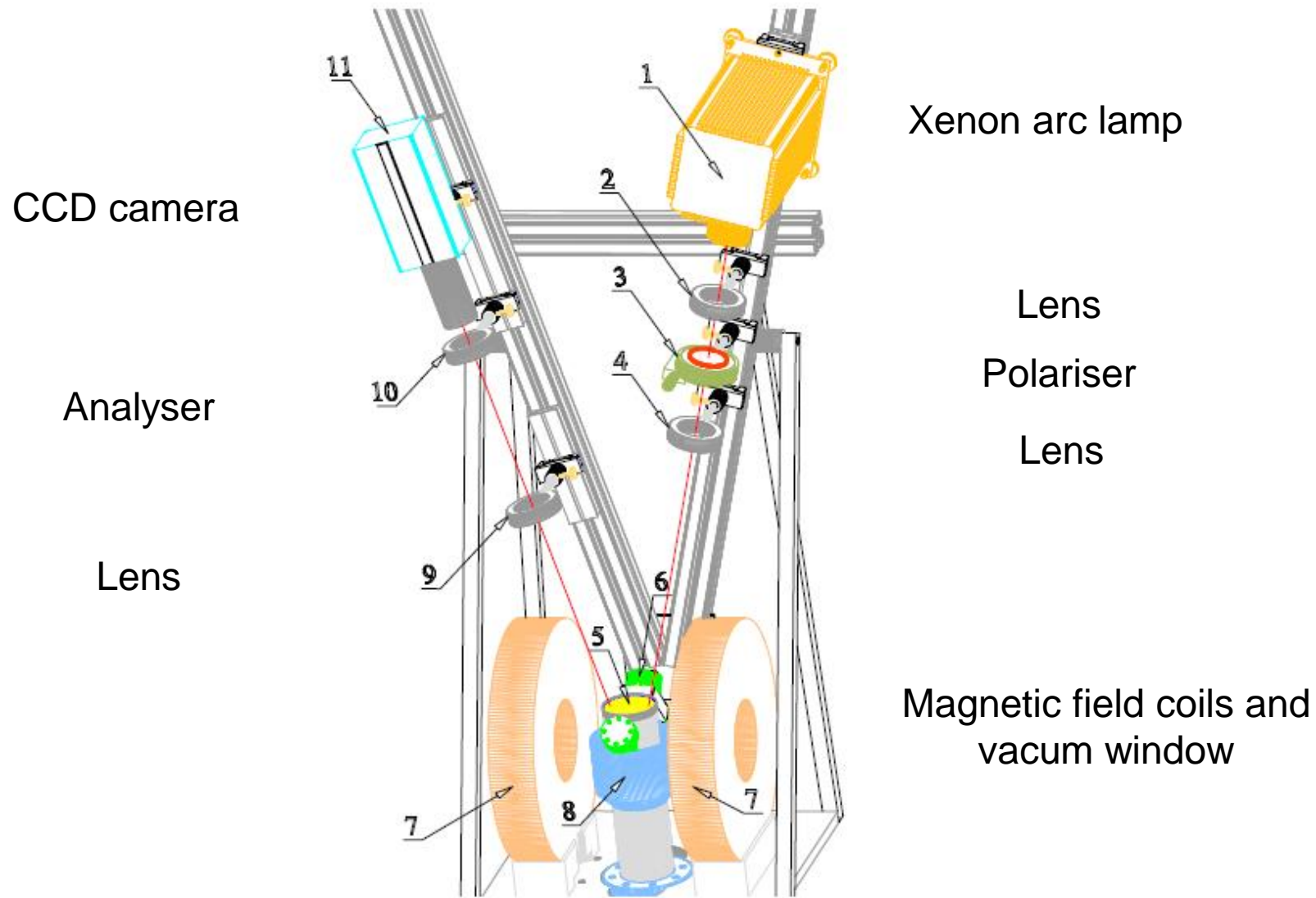


Albrecht et al., JMMM 113 (1992) 207-220

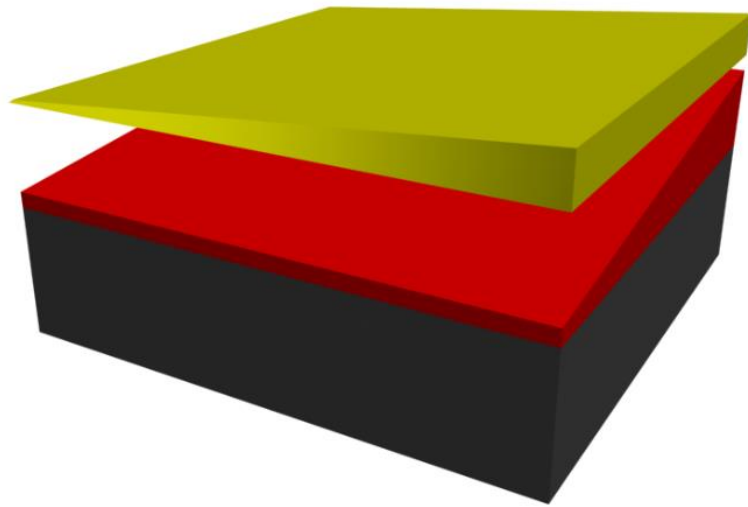


Baek et al., PRB 67, 075401 (2003)

μ MOKE (in-situ) imaging



Co influence on SRT in Fe(110): double wedge samples at AGH



Co wedge 0 - 30 Å

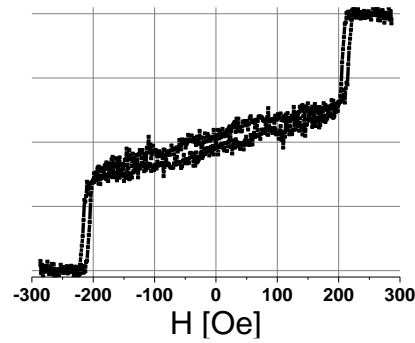
Fe(110) wedge 80 - 300 Å

@ RT, anneal. 400 °C / 8 min

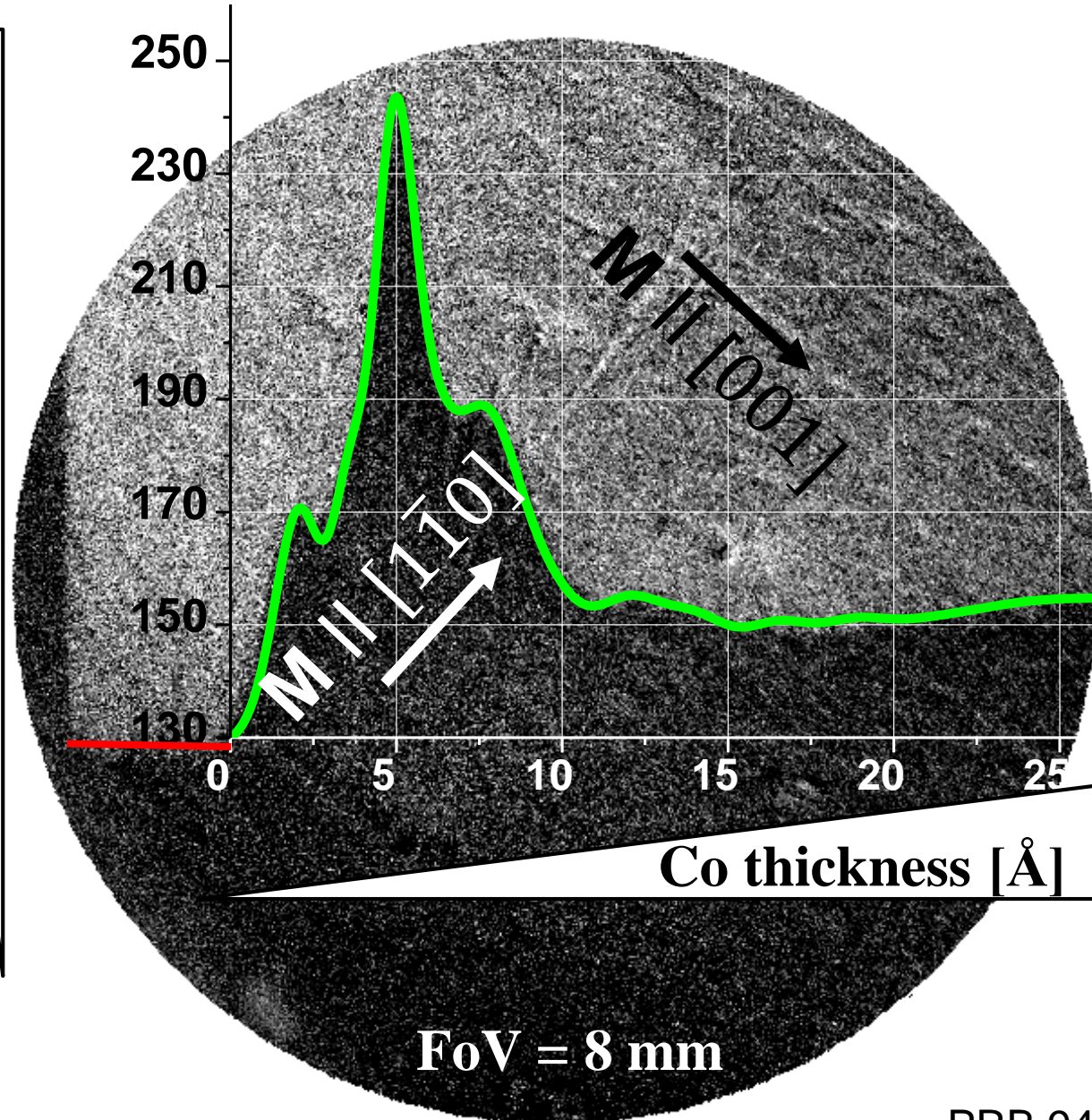
W(110)

Co influence on SRT in Fe(110):

in-situ μ MOKE results



Fe thickness [\AA]



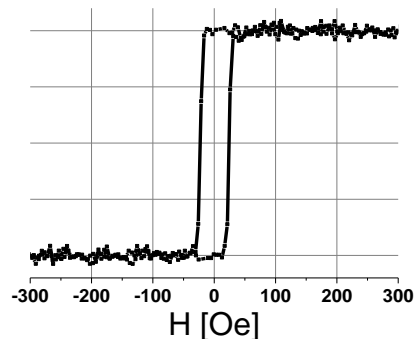
Co thickness [\AA]

FOV = 8 mm

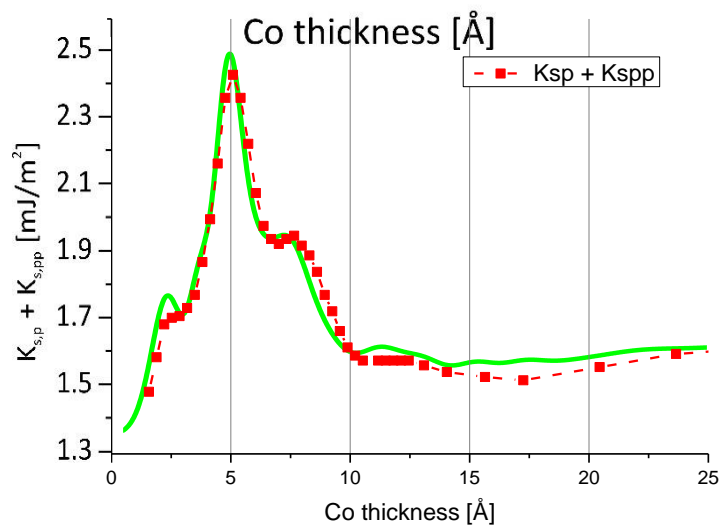
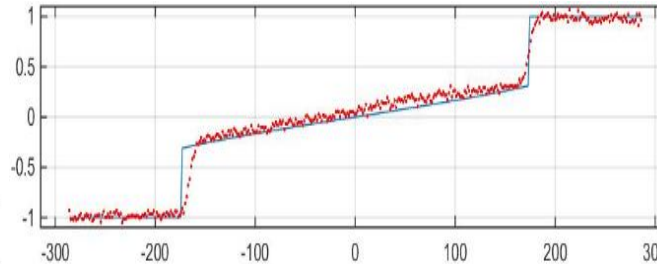
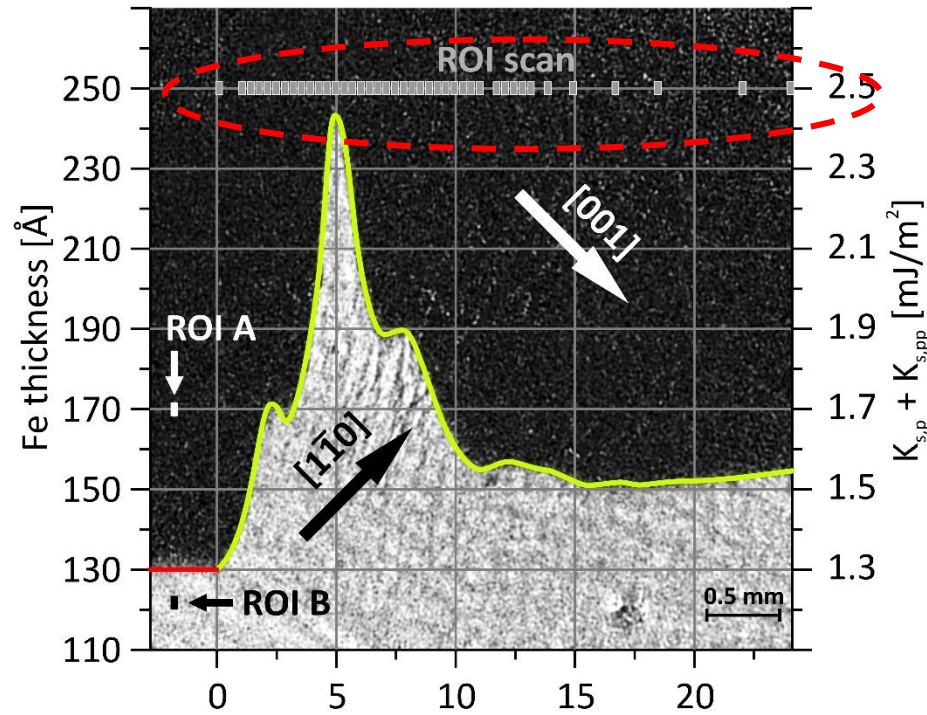
- double-wedged samples
- Fe and Co MBE
- grown on W(110)
- Fe 80 – 300 \AA
- Co 0 – 30 \AA

Remanent MOKE image
B II [1-10]

Critical thickness
Fe/UHV



Magnetic anisotropy at the Co/Fe(110) interface: magnetic hysteresis loops simulations



Model (coherent rotation of magnetization)

Global minimum of energy:
 $E = A \sin^2(\theta) + B \sin^4(\theta) - M_s H \sin(\theta)$

θ - angle between \mathbf{M} and $[001]$

$$A = K_{vp} - K_{sp}/d$$

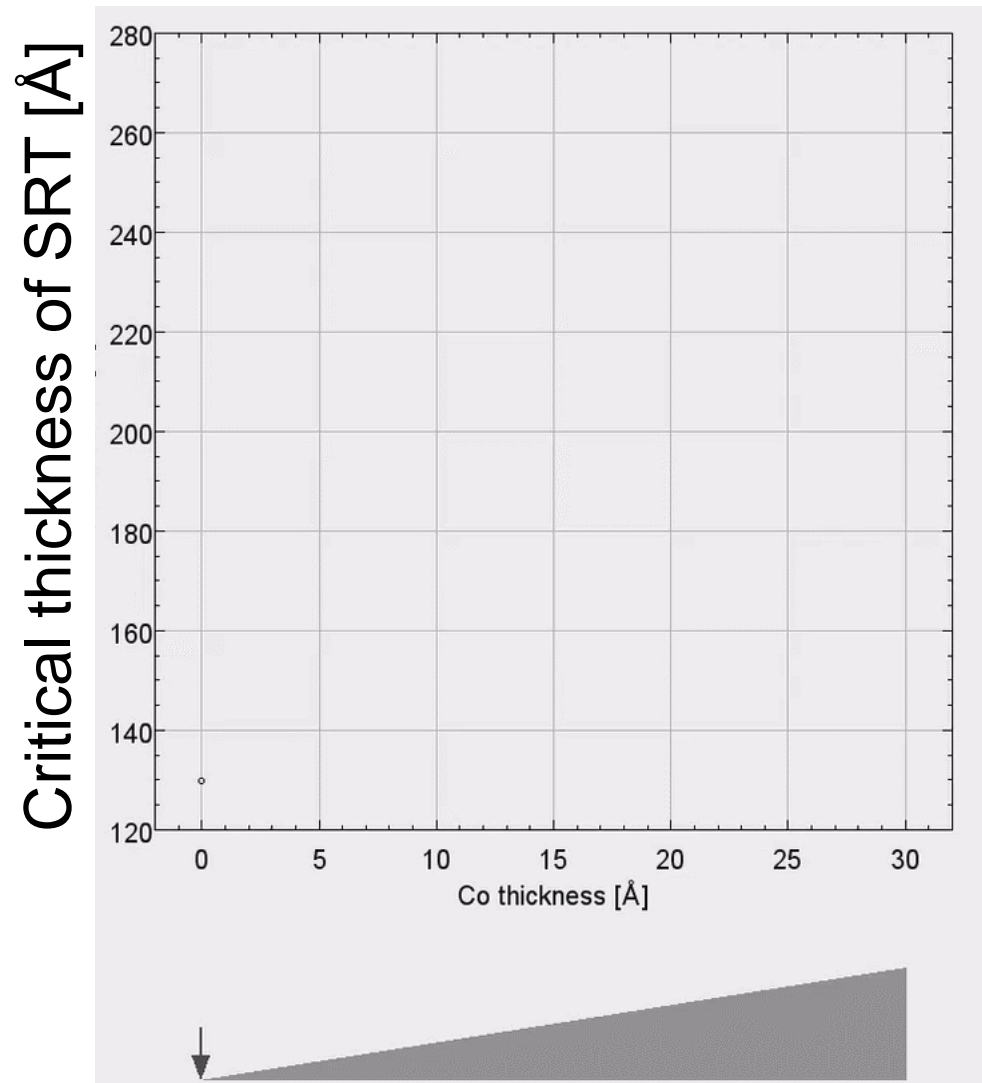
$$B = K_{vpp} - K_{spp}/d$$

$$d_{Fe} = 250 \text{ \AA} = \text{const.}$$

$$K_{vp} = 10.6 \cdot 10^4 \text{ J/m}^3 = \text{const.}$$

$$K_{vpp} = -0.60 \cdot 10^4 \text{ J/m}^3 = \text{const.}$$

Magnetic Anisotropy and SRT in Co/Fe(110)



Oscillations of Ks- morphology of Co film

PHYSICAL REVIEW B

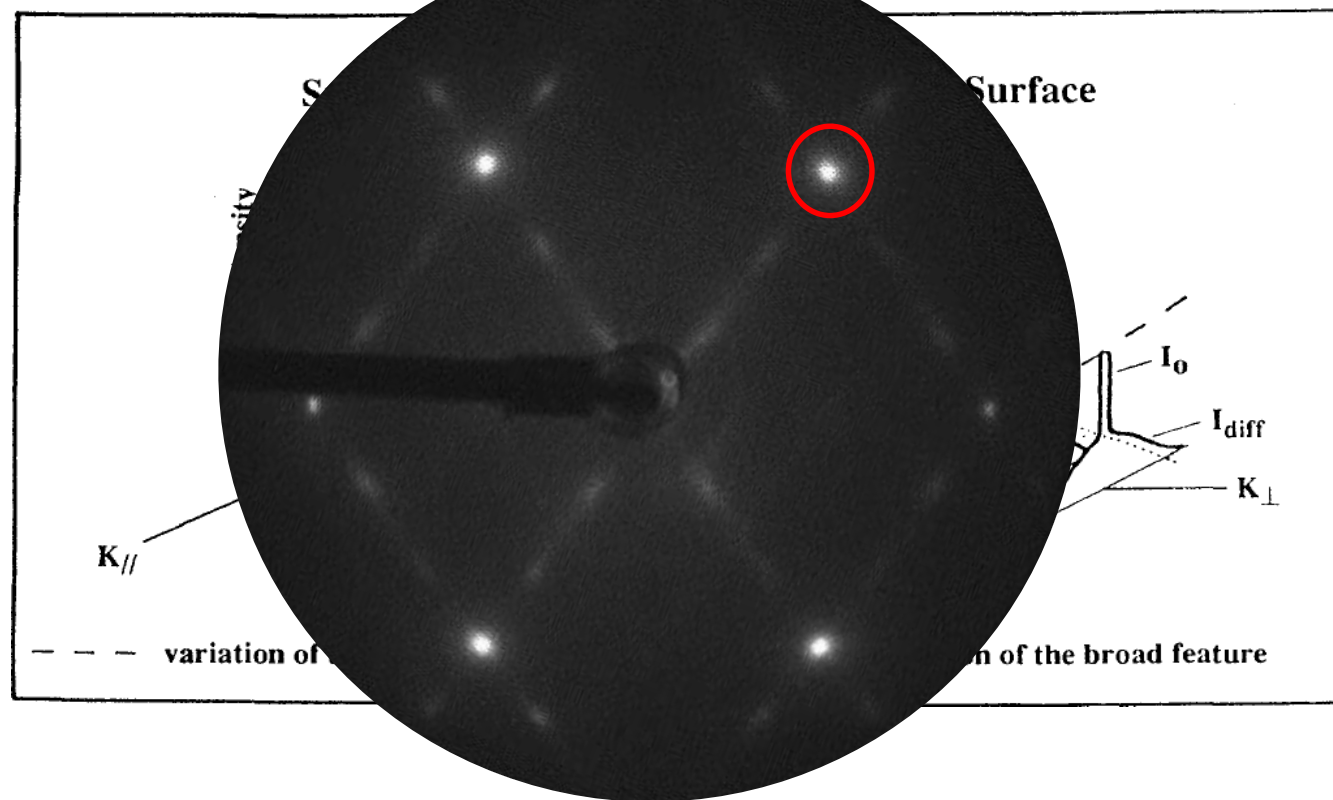
VOLUME 48, NUMBER 19

15 NOVEMBER 1993-I

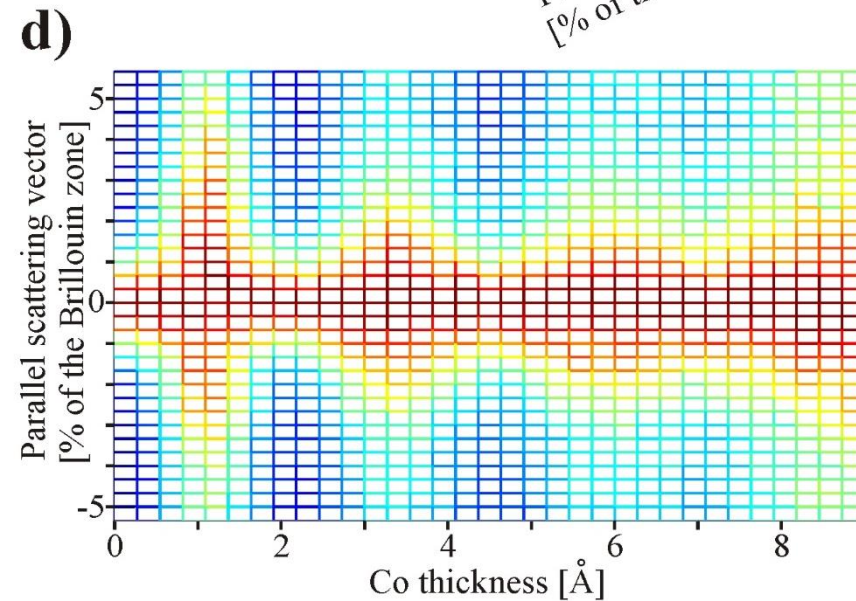
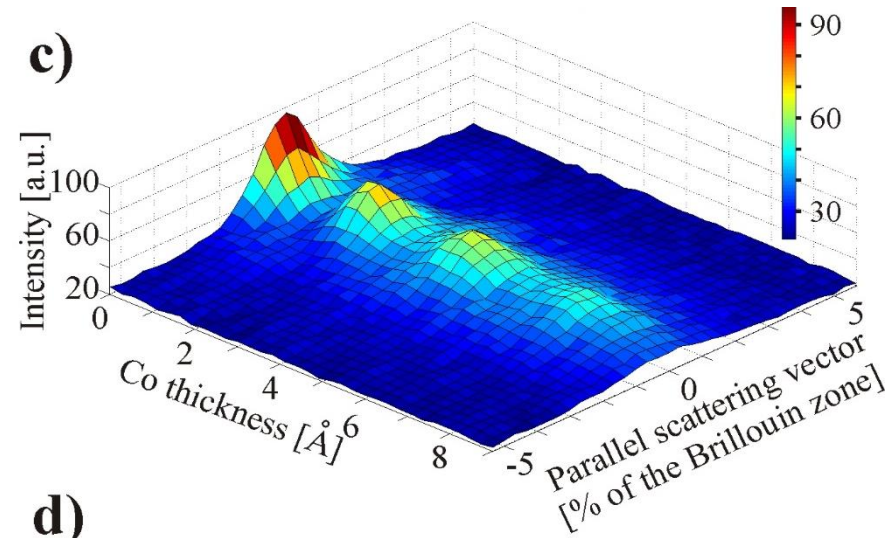
Spot-profile-analyzing LEED study of the epitaxial growth of Fe, Co, and Cu on Cu(100)

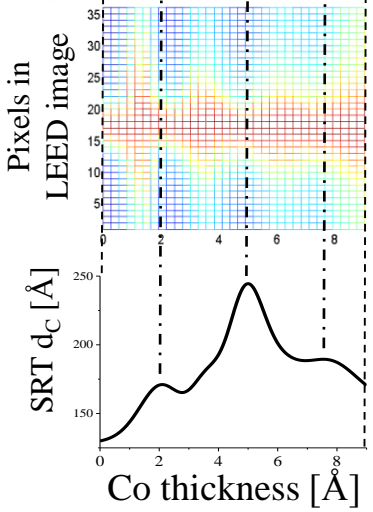
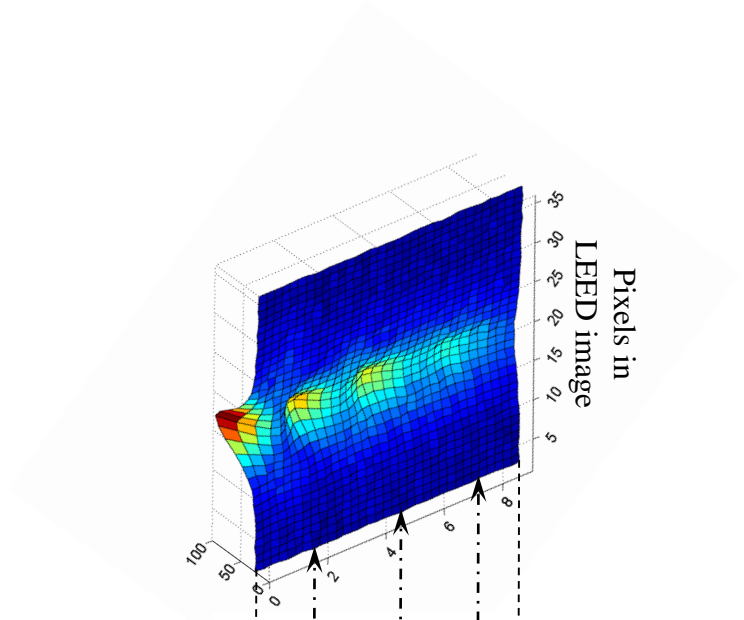
G. L. Nyberg,* M. T. Kief, and W. F. Egelhoff, Jr.

Surface and Microanalysis Science Division, National Standards and Technology, Gaithersburg, Maryland 20899



Co / Fe(110): oscillations of Ks vs oscillations of LEED spot profile

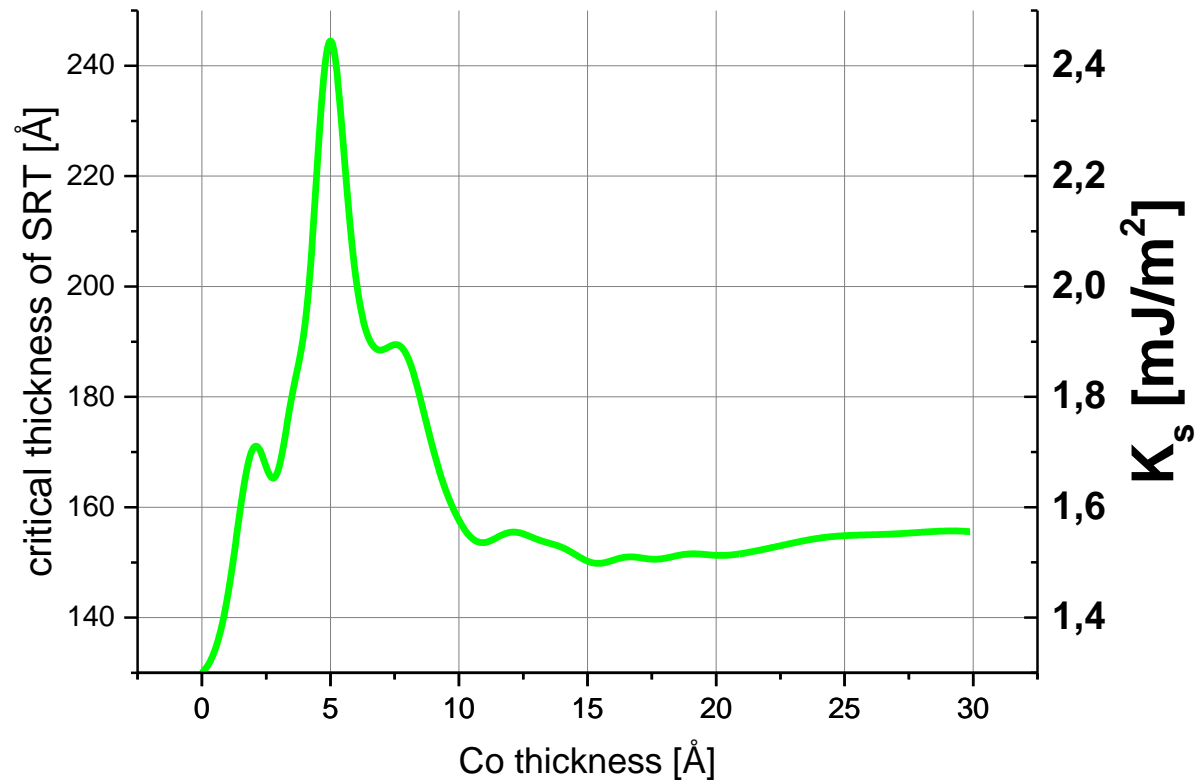


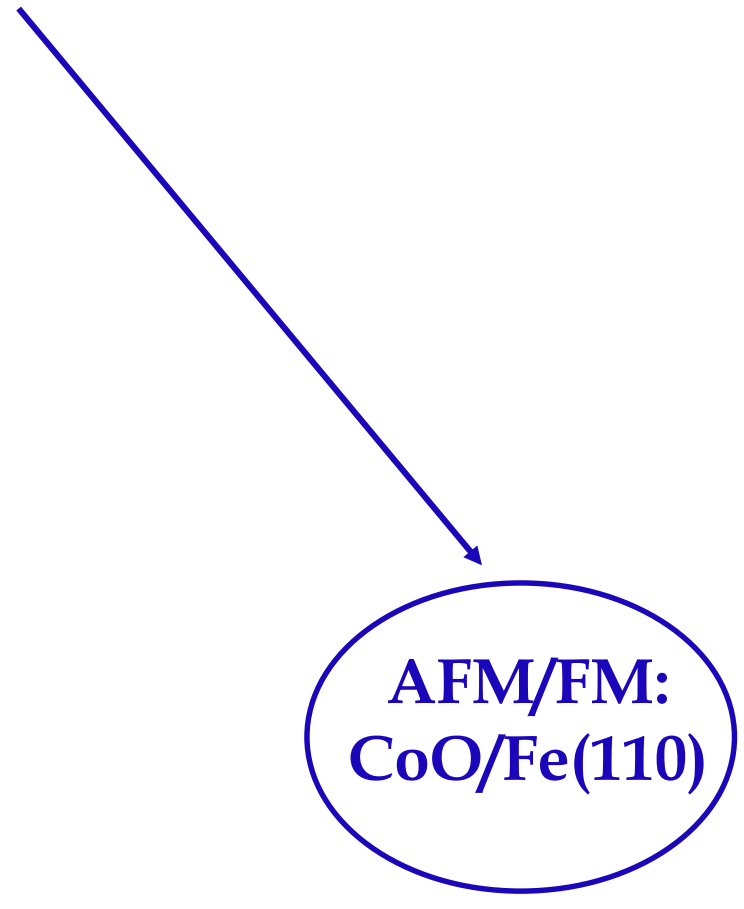
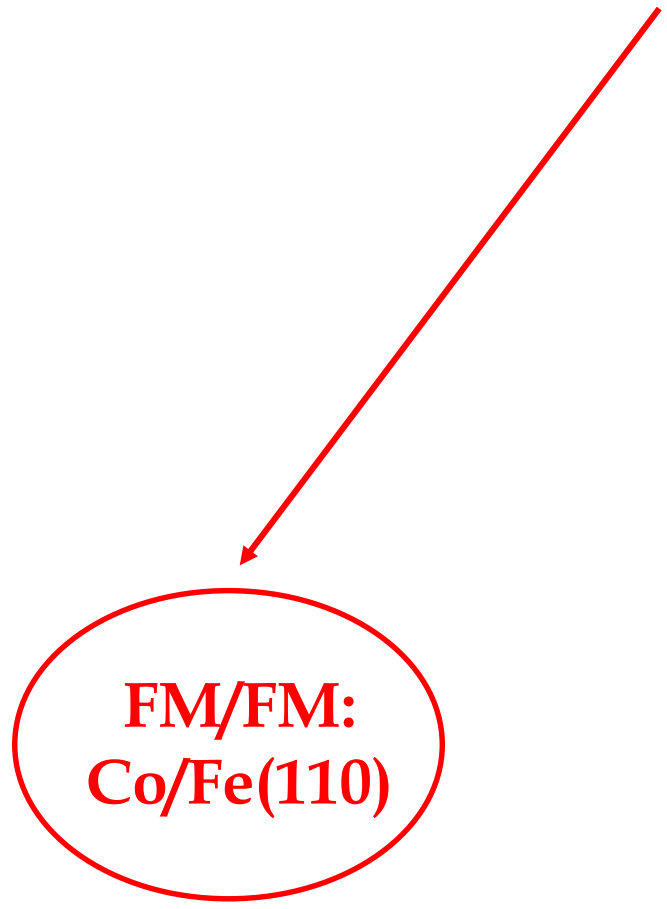


Summary: Co/Fe(110)

→ huge magnetic anisotropy at the Co/Fe(110) interface

→ monolayer (Co) period oscillations of magnetic anisotropy





Discovery of *exchange bias*

Phys. Rev., 105, 904 (1957)

New Magnetic Anisotropy

W. H. MEIKLEJOHN AND C. P. BEAN

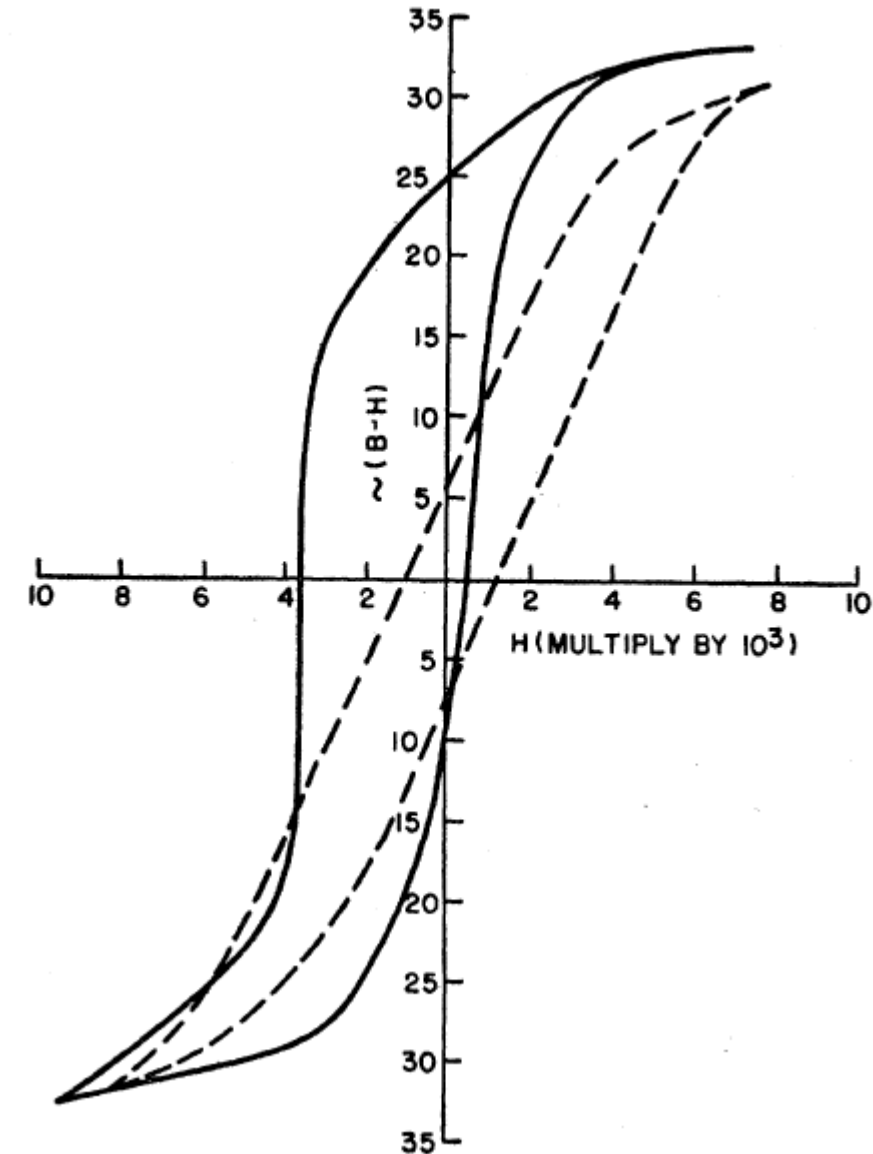
General Electric Research Laboratory, Schenectady, New York

(Received March 7, 1956)

Co particles embedded in their native
antiferromagnetic oxide CoO

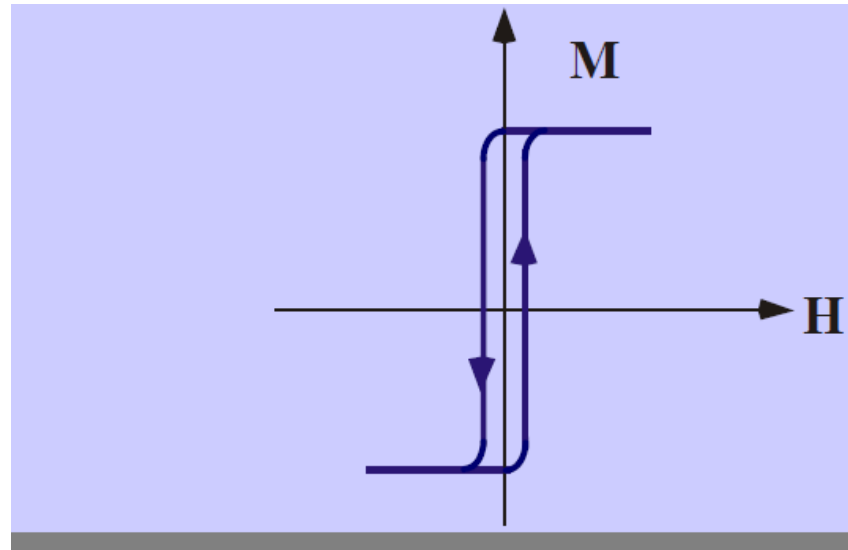
W. H. Meiklejohn and C. P. Bean, Phys. Rev. 102, 1413 (1956).

W. H. Meiklejohn and C. P. Bean, Phys. Rev. 105, 904 (1957).



Introduction

exchange bias

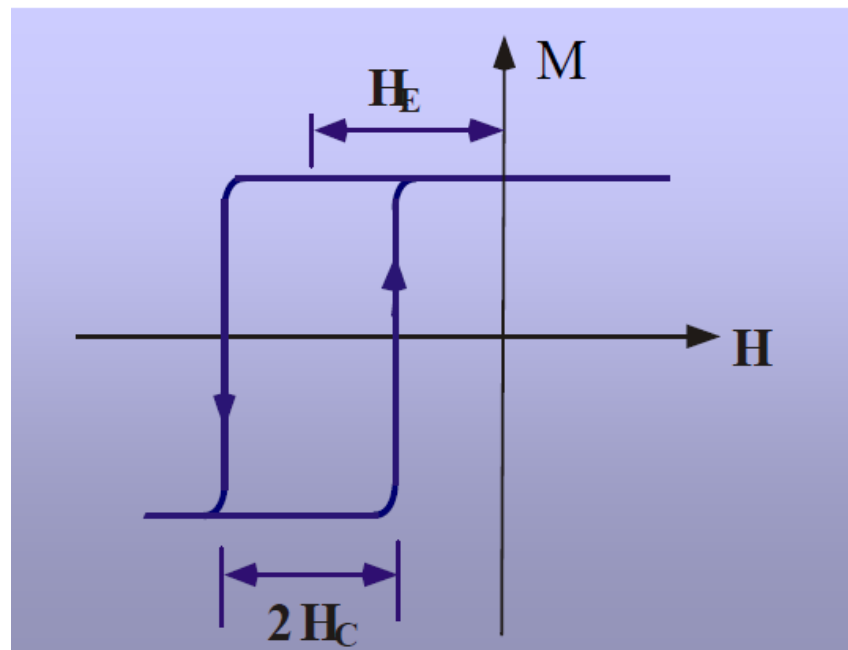


Free FM

Exchange field $H_E=0$

Small coercivity H_C

Symmetric



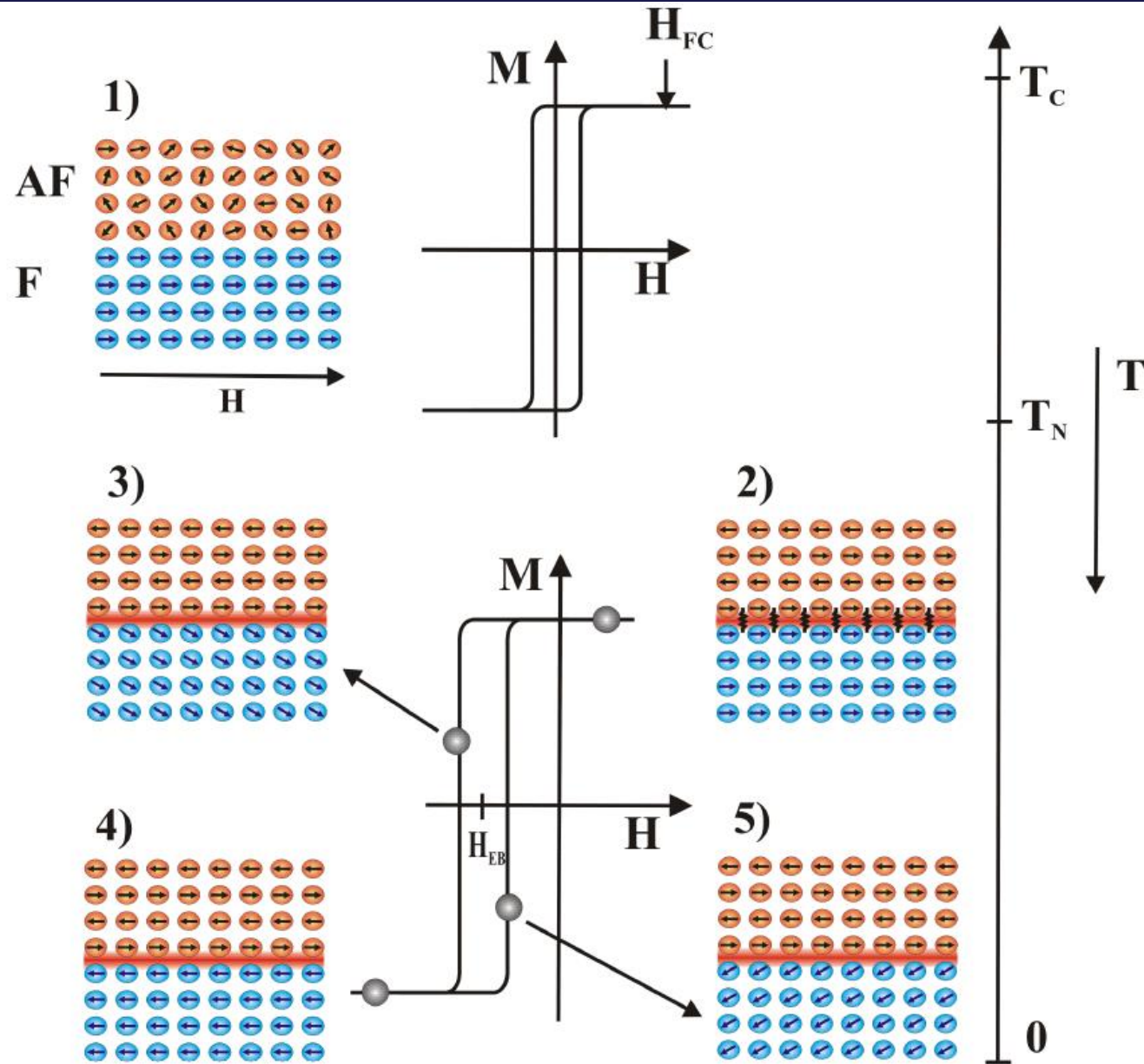
Pinned FM

Large H_E

Large H_C

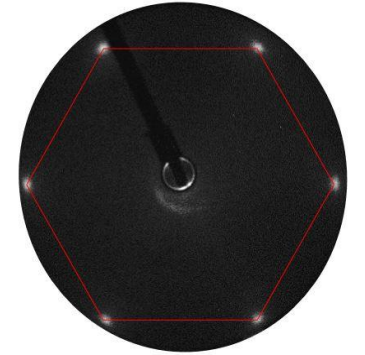
Introduction

exchange bias

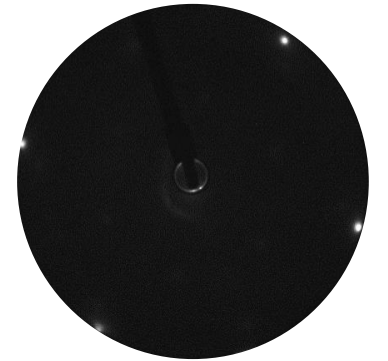
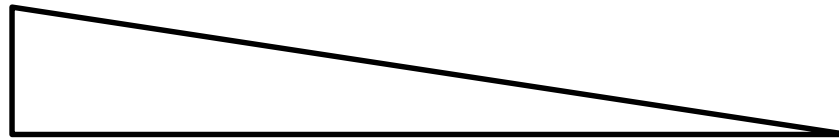


CoO/Fe(110) sample(s)

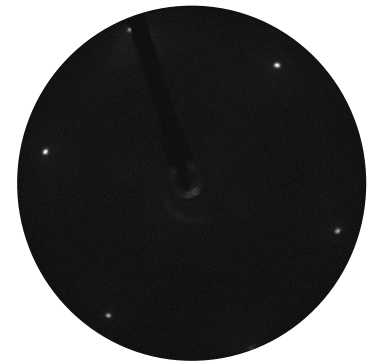
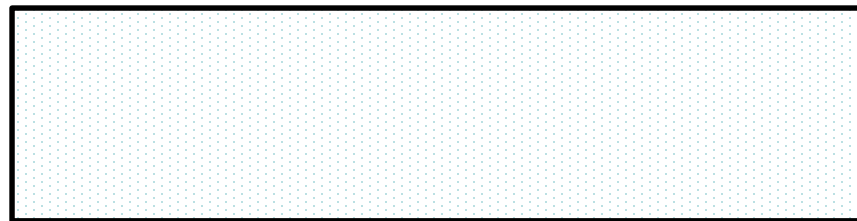
CoO(111): 90 Å



Fe(110) wedge: 80 - 300 Å

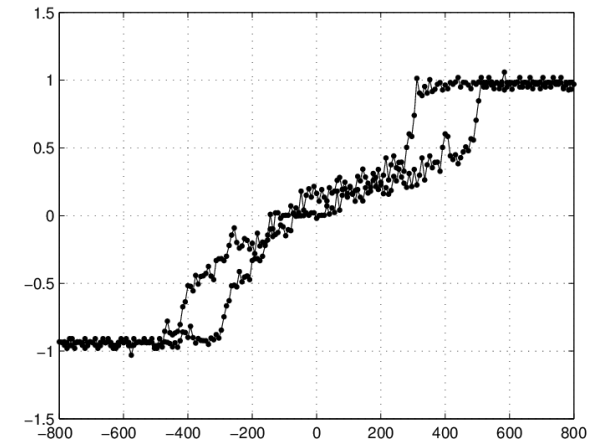
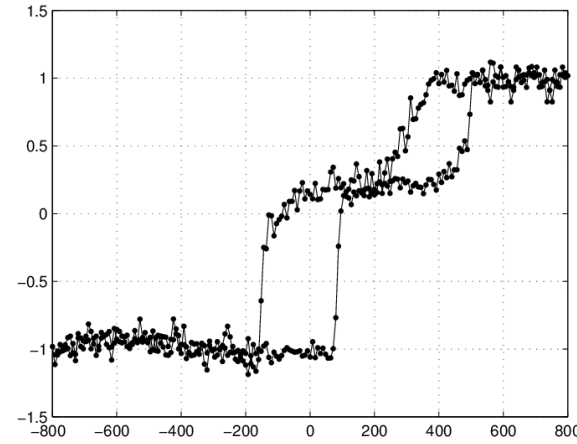
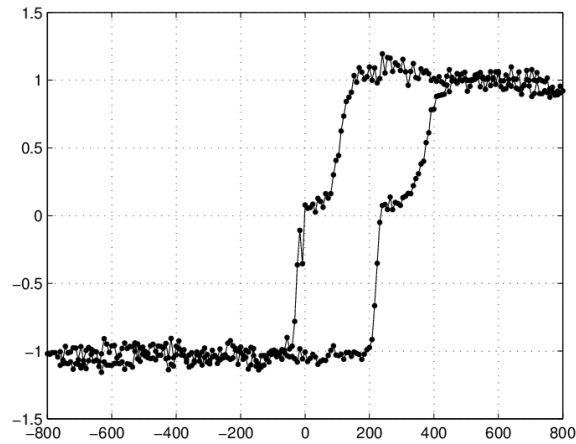
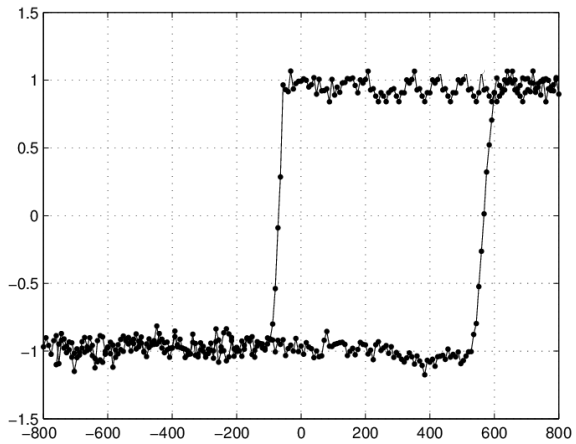


substrate: W(110) single crystal



CoO(111) on Fe(110): evolution of magnetic hysteresis loops @ 183 K

B II [1-10]

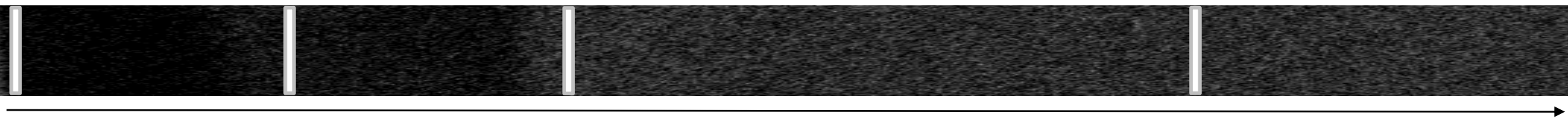


$d_{\text{Fe}} = 80 \text{ \AA}$

$d_{\text{Fe}} = 110 \text{ \AA}$

$d_{\text{Fe}} = 140 \text{ \AA}$

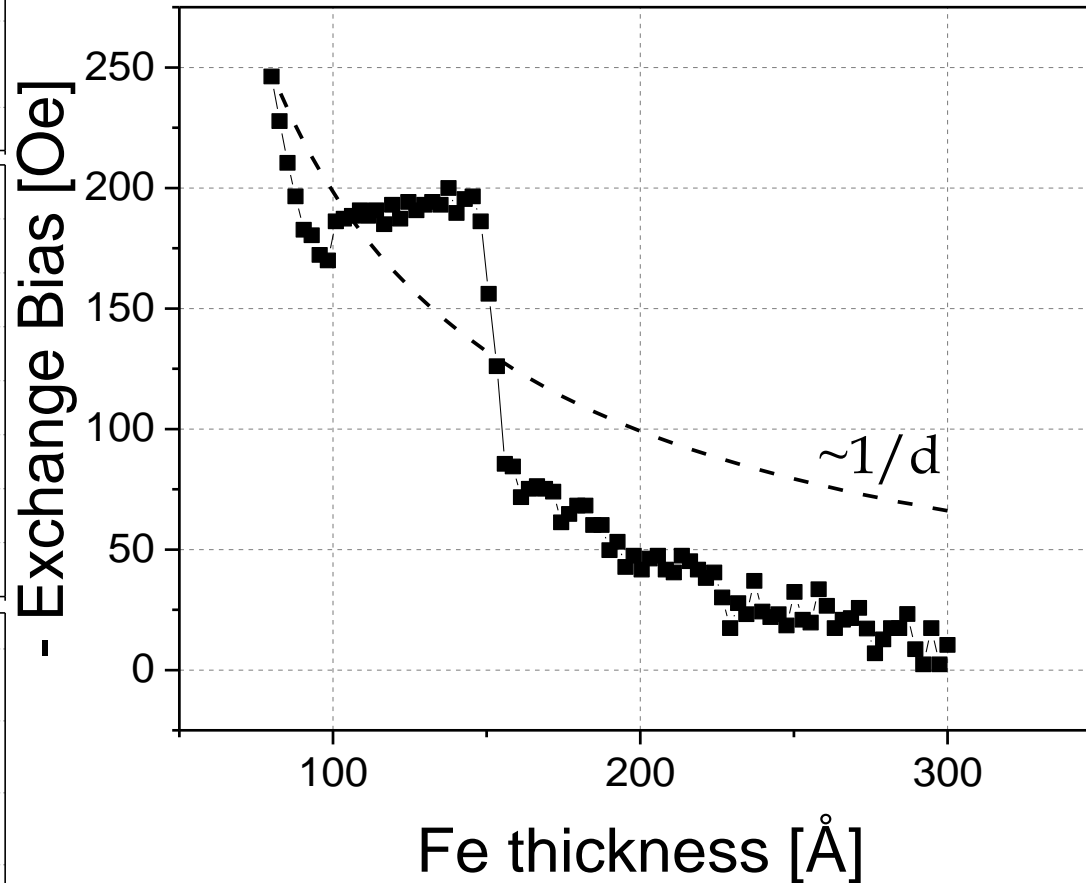
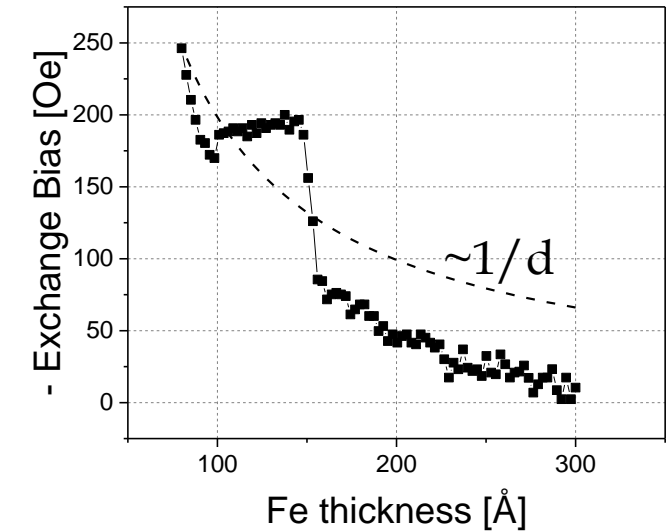
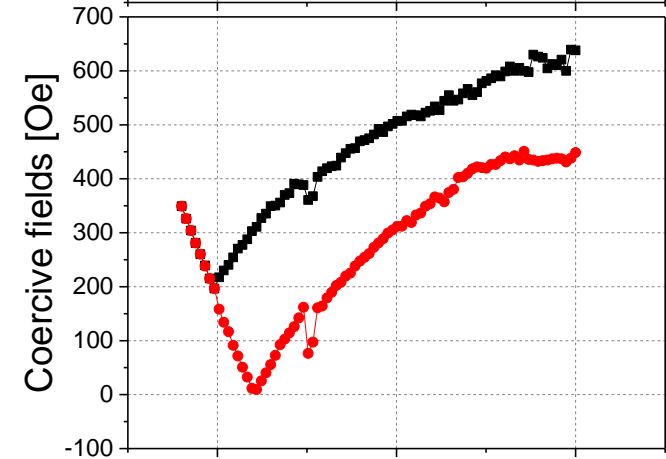
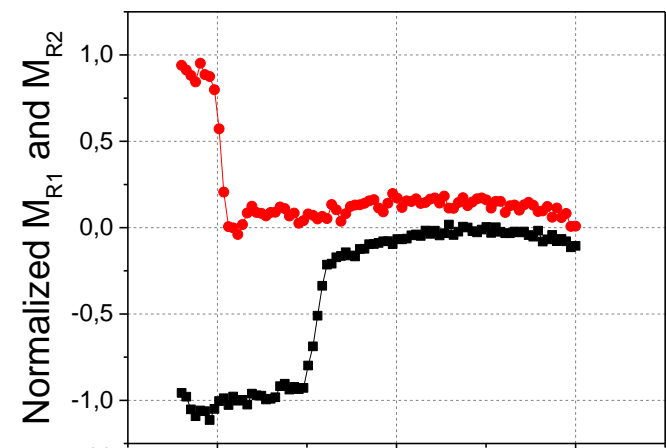
$d_{\text{Fe}} = 200 \text{ \AA}$



SRT in CoO/Fe
EB in CoO/Fe

Fe thickness

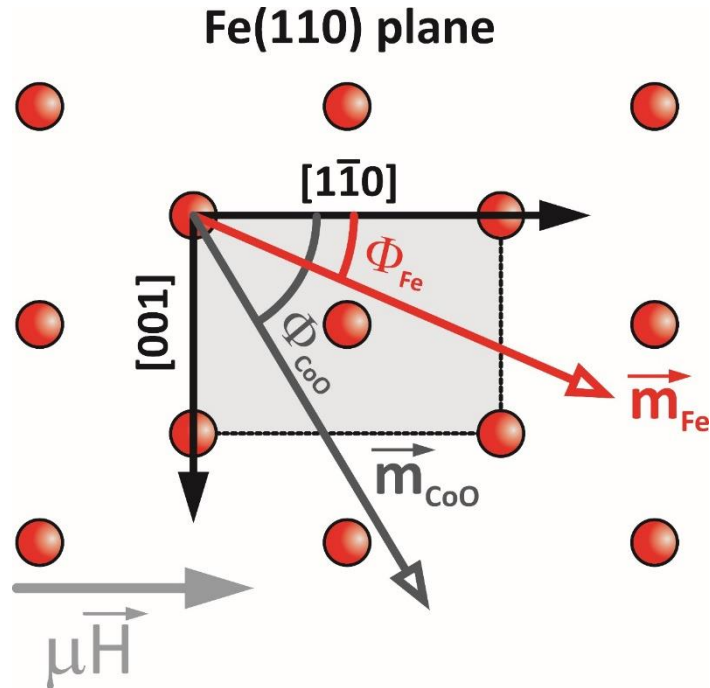
CoO(111) on Fe(110): evolution of magnetic hysteresis loops



- Non-monotonous H_{eb} vs d dependence
- Not proportional to $1/d$

CoO(111) on Fe(110): simulations of magnetic hysteresis loops

$$G(\mathbf{d}_{\text{Fe}}, \Phi_{\text{Fe}}, \Phi_{\text{CoO}}) = \mathbf{E}_{\text{CoO}}(\Phi_{\text{CoO}}, \mathbf{d}_{\text{Fe}}) + \mathbf{E}_{\text{CoO-Fe}}(\Phi_{\text{CoO}}, \Phi_{\text{Fe}}) + \mathbf{E}_{\text{Fe}}(\Phi_{\text{Fe}}, \mathbf{d}_{\text{Fe}}) + \mathbf{E}_{\text{H}}$$



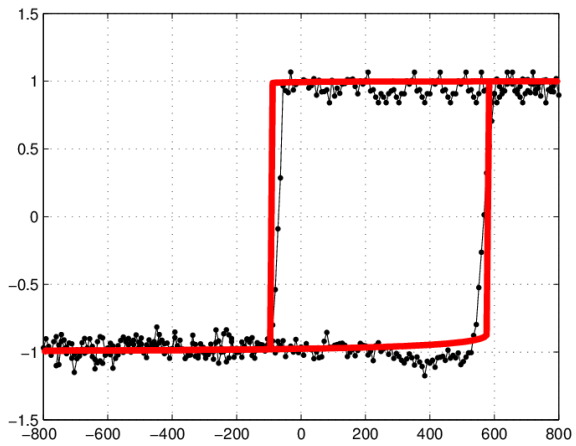
$$\mathbf{E}_{\text{CoO-Fe}}(\Phi_{\text{CoO}}, \Phi_{\text{Fe}}) = -K_{\text{EB}}/d_{\text{Fe}} \cos(\Phi_{\text{Fe}} - \Phi_{\text{CoO}})$$

$$\mathbf{E}_{\text{Fe}}(\Phi_{\text{Fe}}, \mathbf{d}_{\text{Fe}}) = A \cos^2(\Phi_{\text{Fe}}) + B \cos^4(\Phi_{\text{Fe}})$$

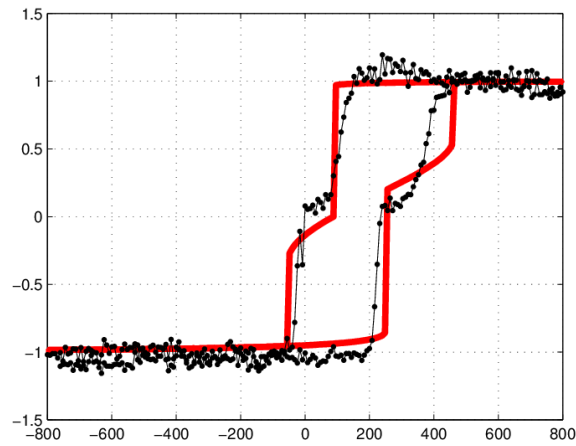
$$\mathbf{E}_{\text{H}} = -M_s H \cos(\Phi_{\text{Fe}})$$

$G(\Phi_{\text{Fe}})$ minimization for each value of $H \rightarrow$ simulation of hysteresis curve

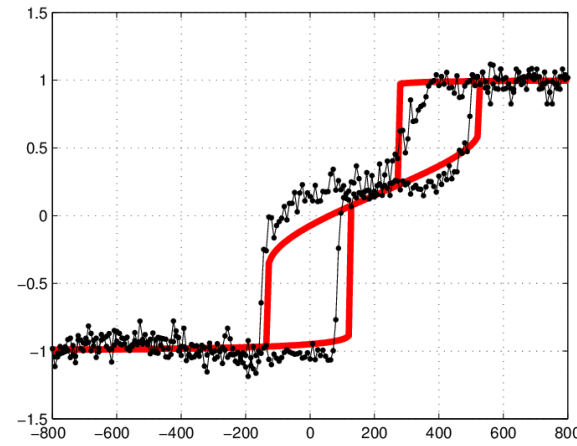
CoO(111) on Fe(110): simulations vs MOKE results



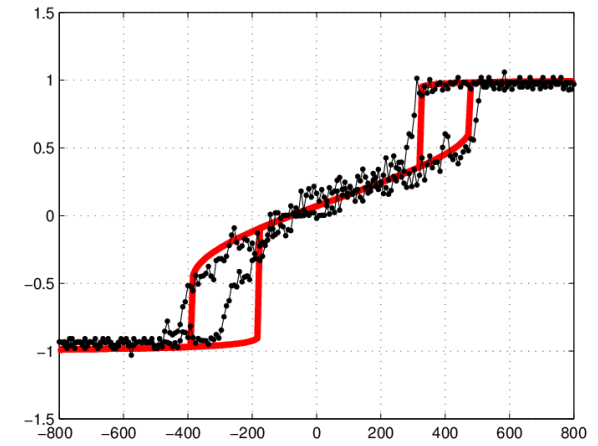
$$d_{\text{Fe}} = 80 \text{ \AA}$$



$$d_{\text{Fe}} = 110 \text{ \AA}$$

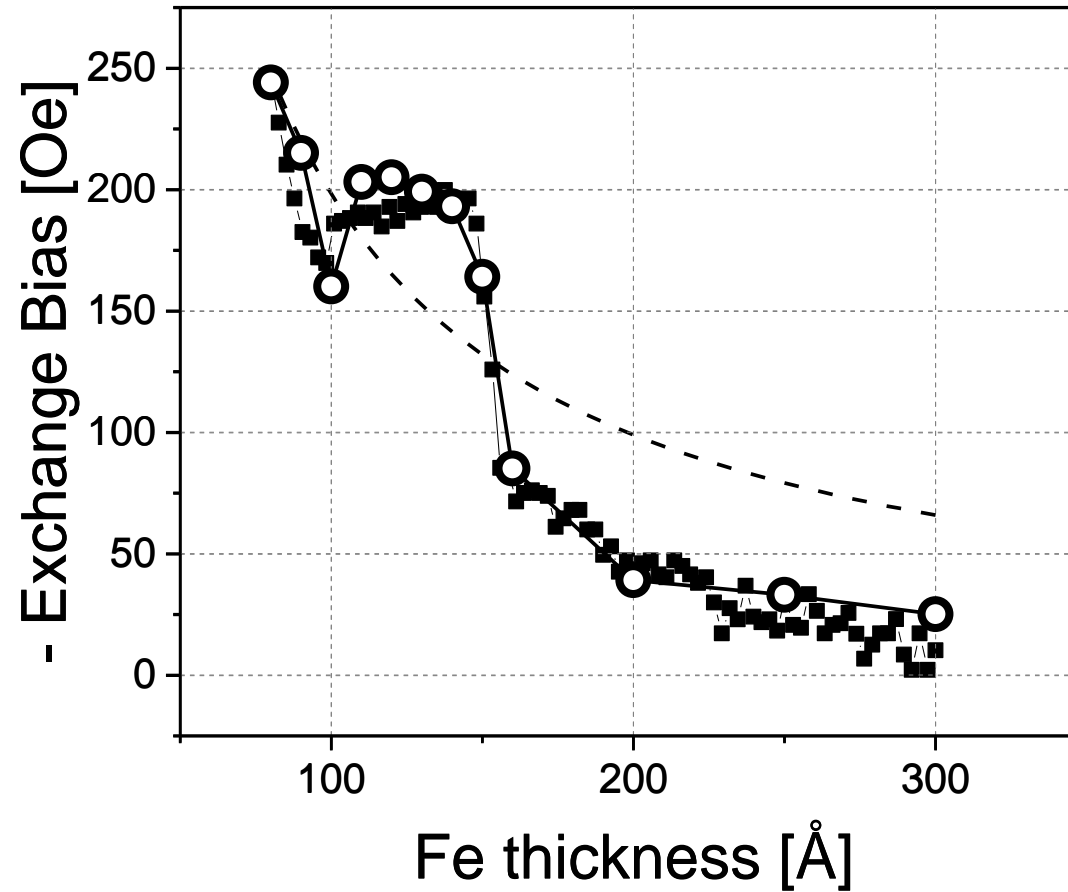
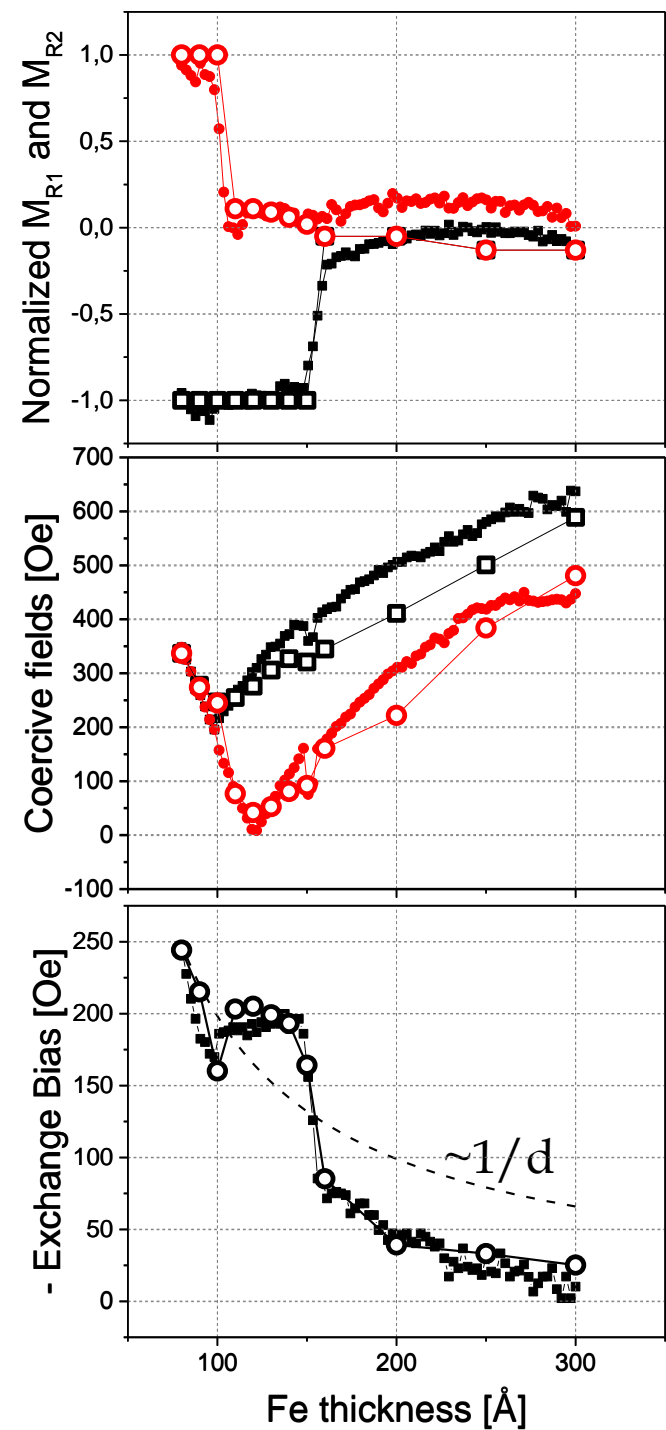


$$d_{\text{Fe}} = 140 \text{ \AA}$$

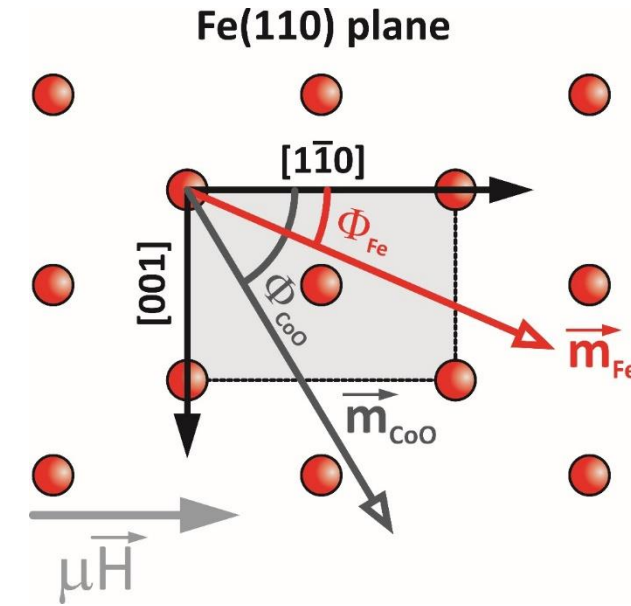
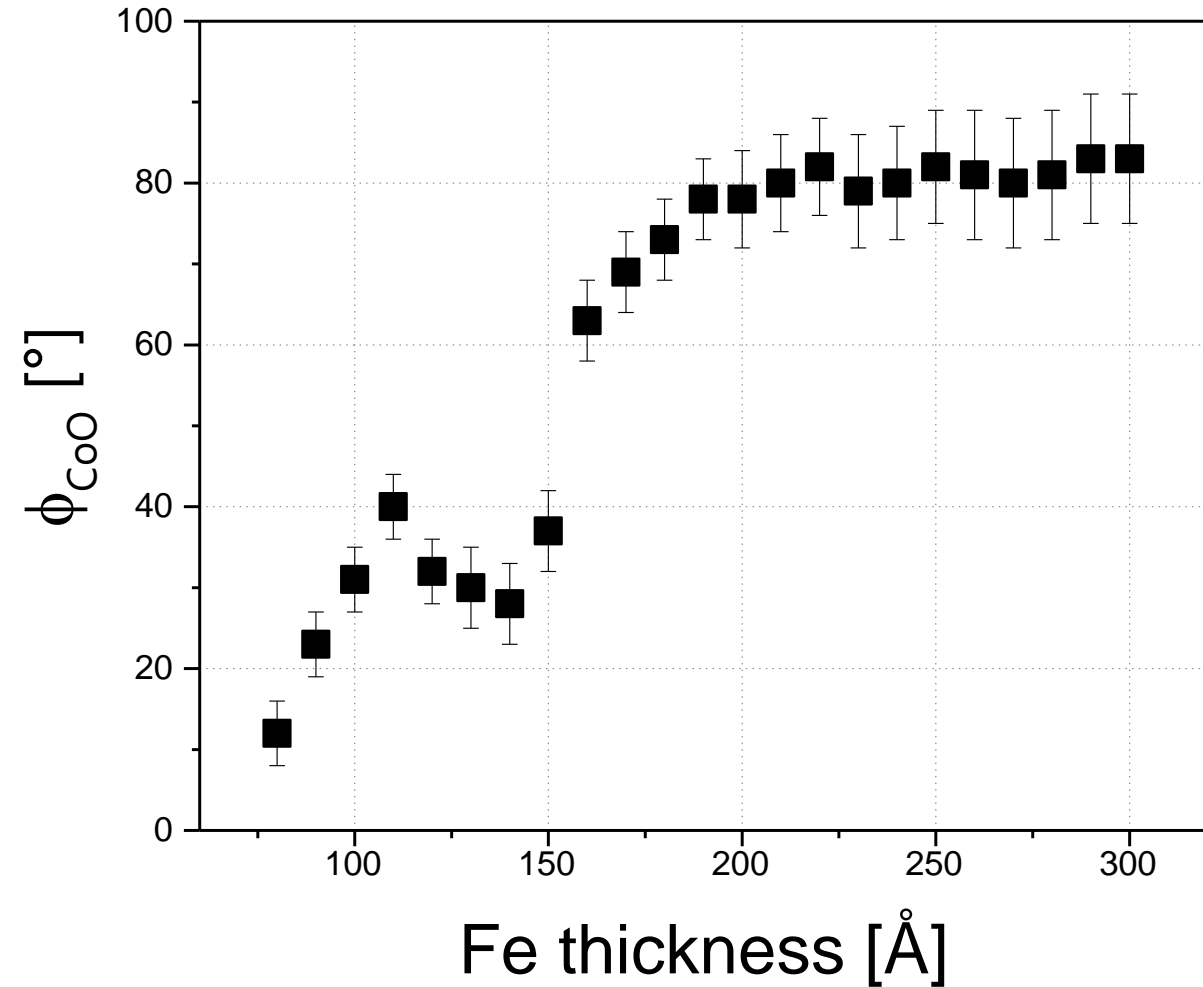


$$d_{\text{Fe}} = 200 \text{ \AA}$$

CoO(111) on Fe(110): simulations vs MOKE results



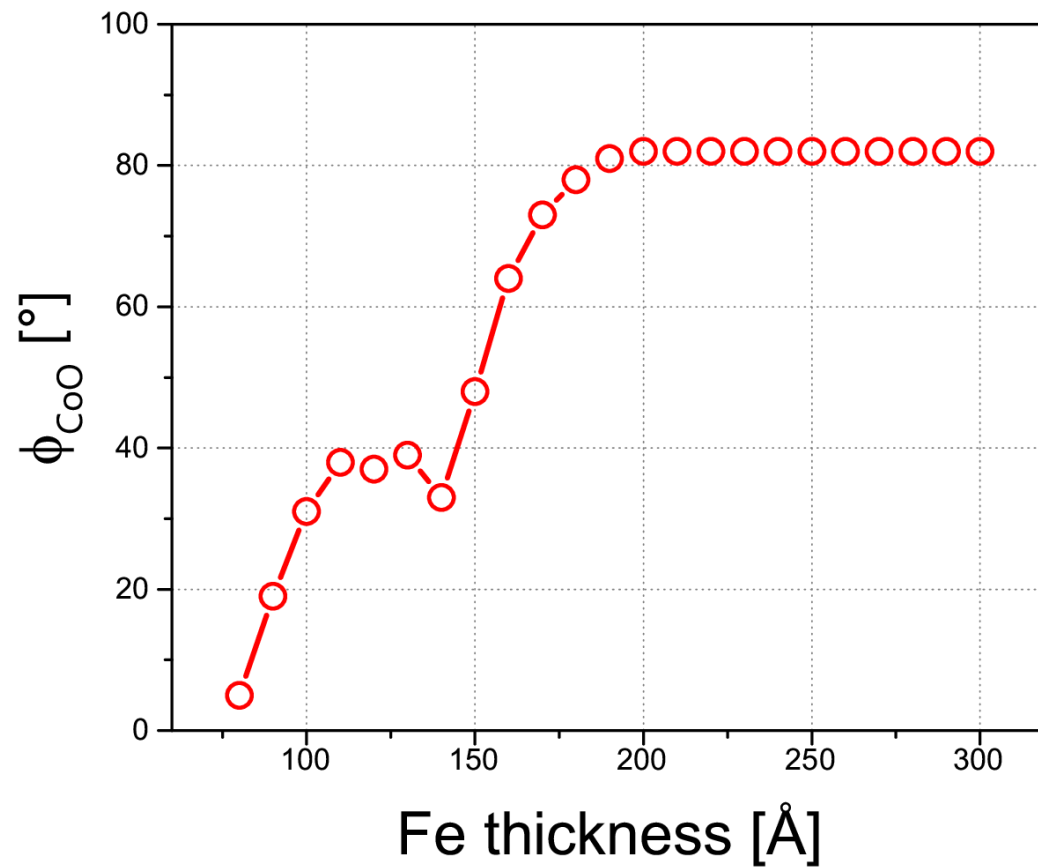
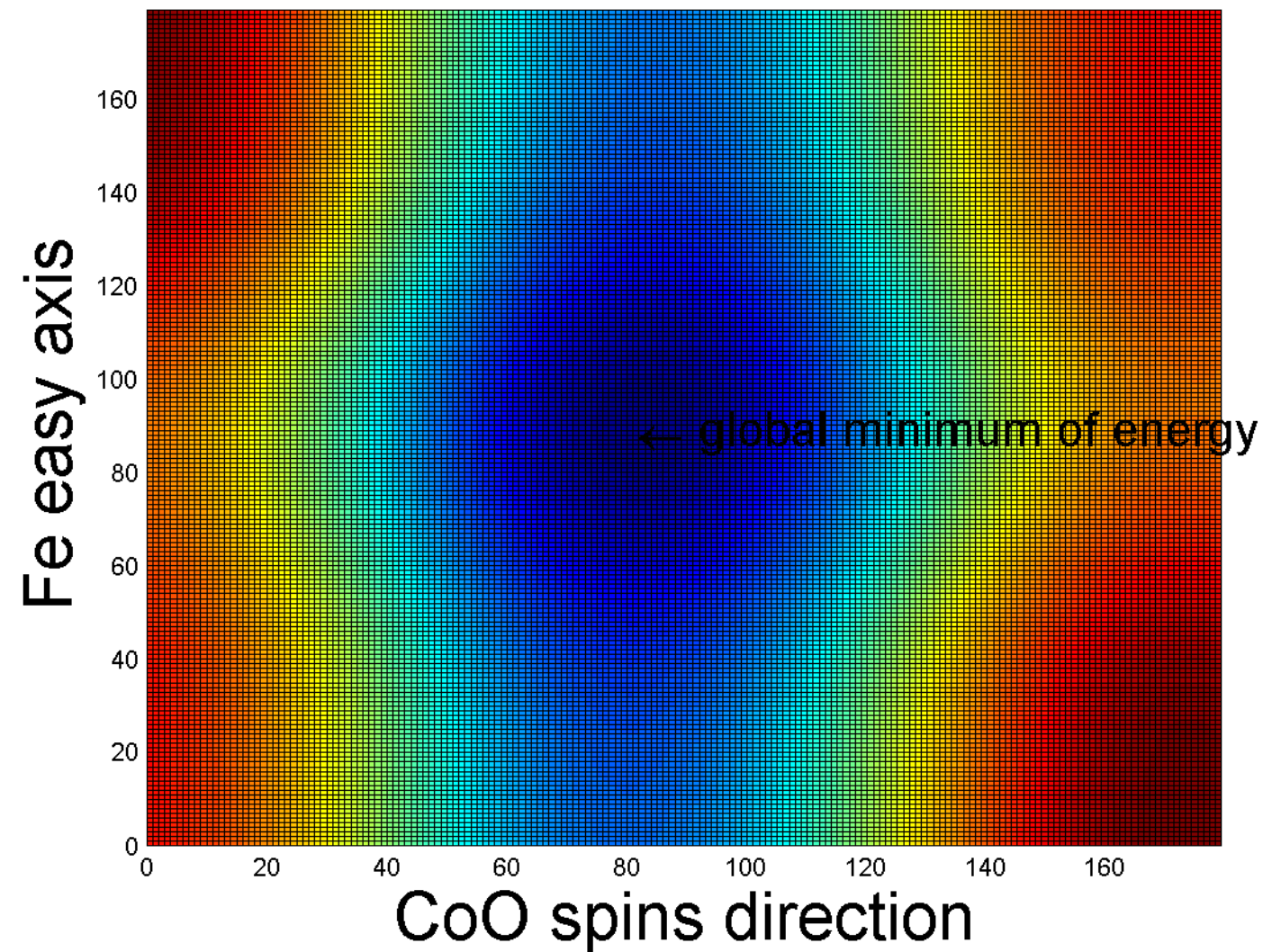
SRT in CoO(111)



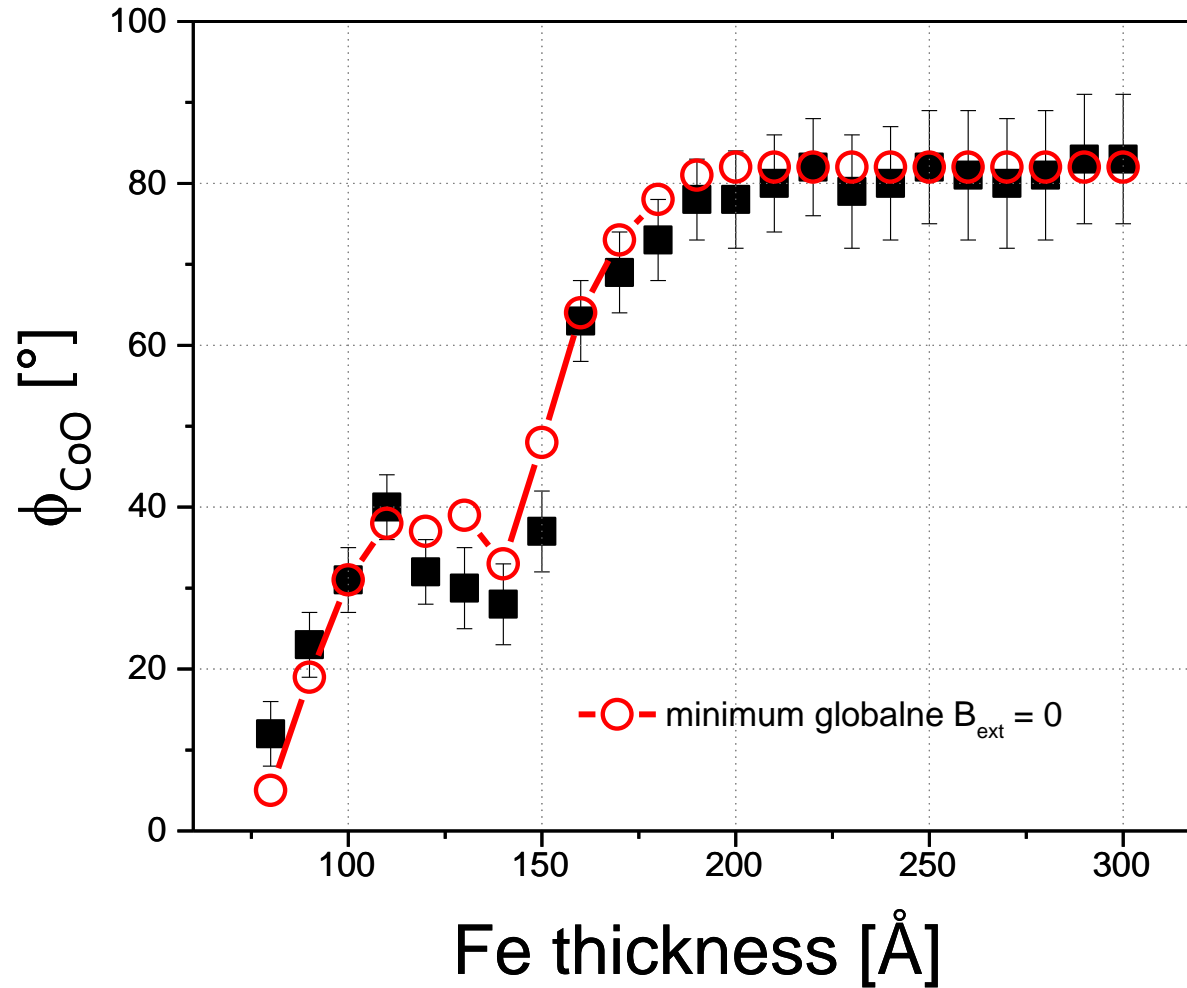
- Fe acts as a local magnetic field when passing T_N
- Fe imprints magnetic anisotropy in CoO
- SRT in Fe drives SRT in CoO

Mechanism of SRT in CoO(111): simulations

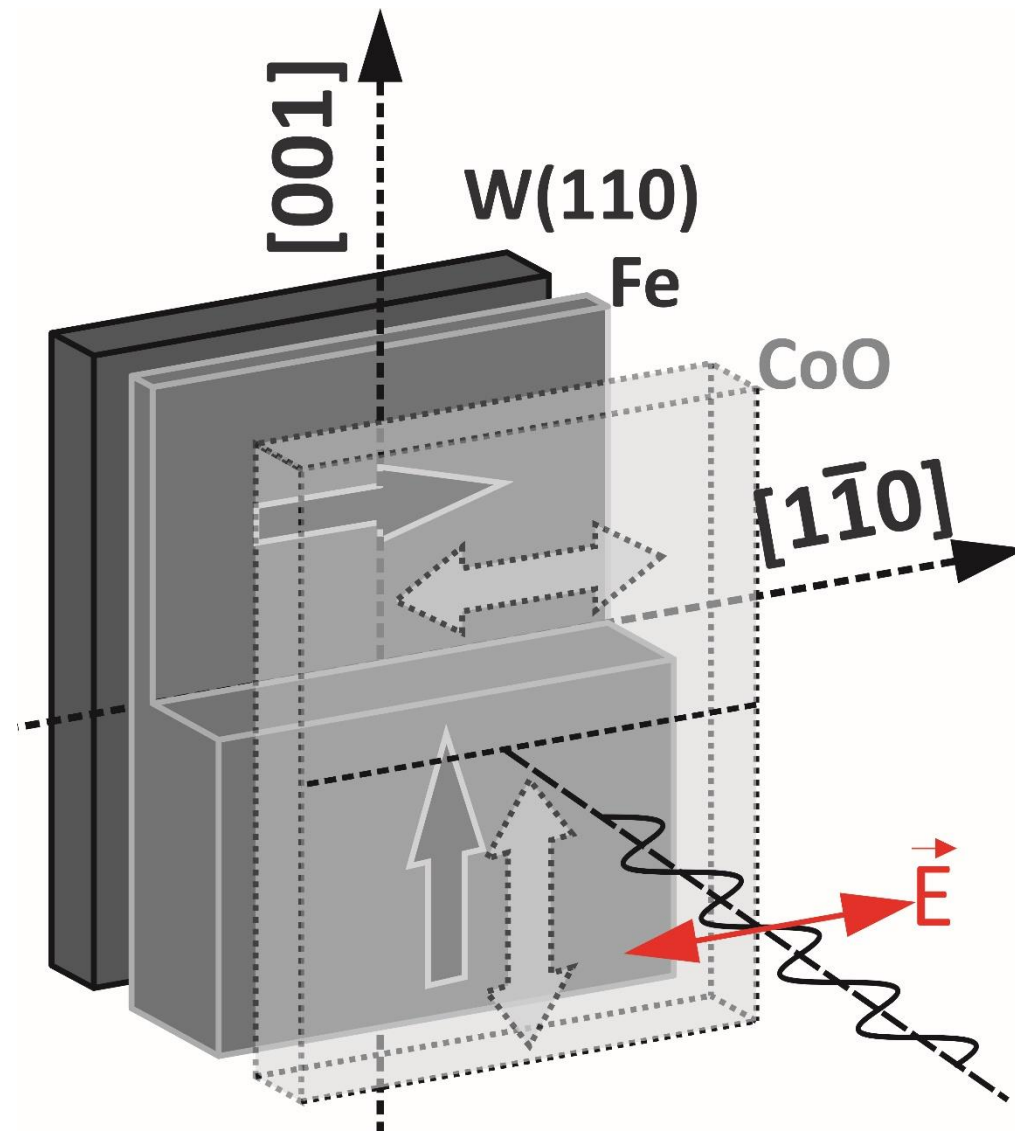
300



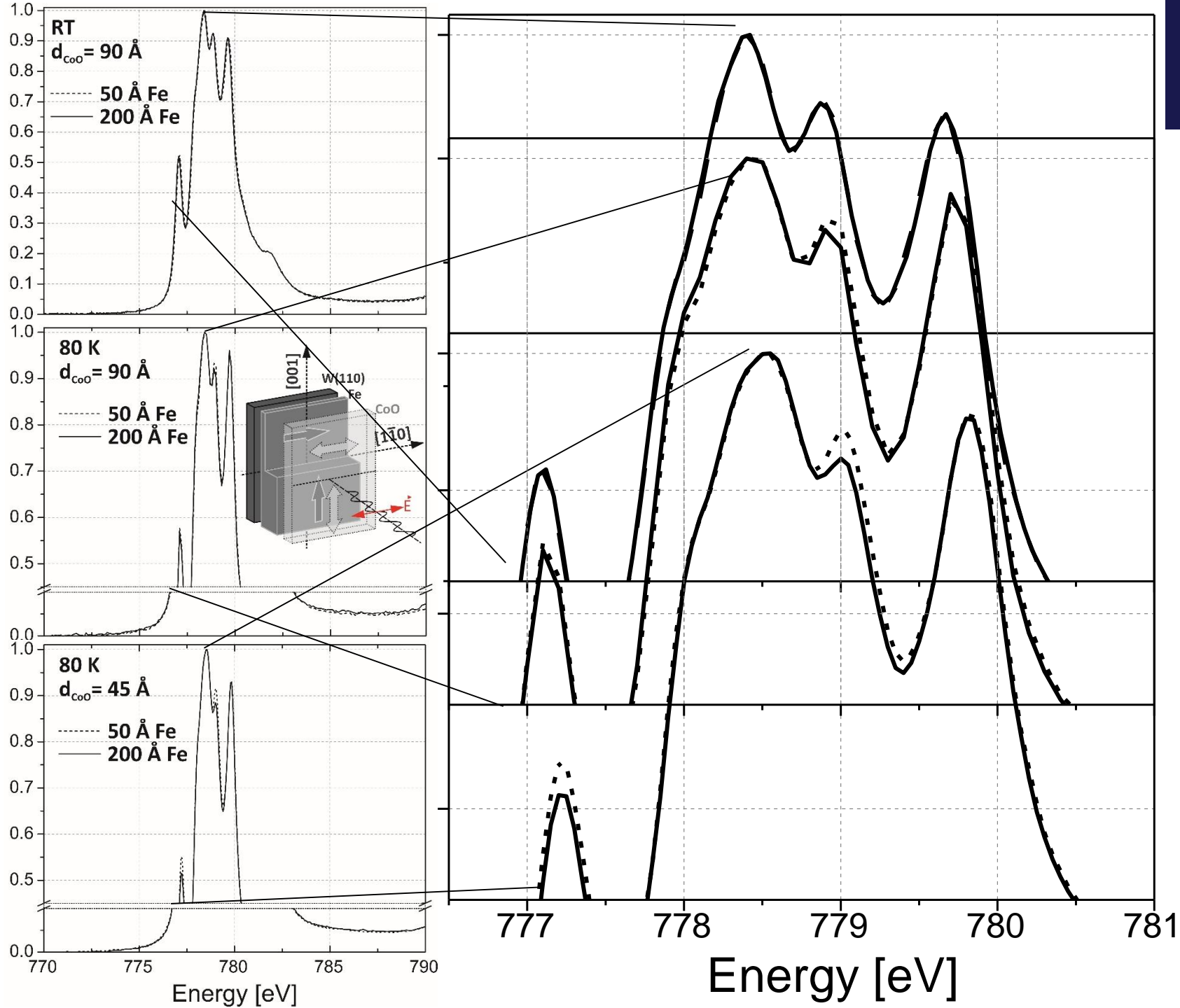
Origin of SRT in CoO(111)



Looking for direct evidence of SRT in CoO: XMLD @ Solaris, XAS end-station



Direct evidence of SRT in CoO: XMLD



→ rotation of CoO spins
from Fe[1-10] to
Fe[001]

following the XMLD analysis by Li et al., *Phys. Rev. B* **91**, 104424 (2015)